# Nanoindentation System for Material Properties Identification

Adrian Wei Hong, Teo, Gik Hong, Yeap, and Wei Jie, Loo

Abstract—This paper proposed to design a nanoindentation system with the intention to identify material properties without spalling. The system is designed to perform simulation based on the load-depth curve data collected from NanoTest<sup>TM</sup>. The hardness results are compared with Brinell hardness test and NanoTest<sup>TM</sup> for the same materials; i.e. brass, mild steel, aluminium and copper. Oliver-Pharr and Joslin-Oliver methods are selected to measure the material properties. Both selected methods require indentation load, impression area and depth to fulfil the material properties calculation while these signals are collected through a displacement sensor and an actuator. The results collected indicate that the spall of material rate, which can be reduced by decreasing the indentation load while maintaining the indentation depth at a longer dwell time. The theoretical simulation result of Joslin-Oliver method which neglects substrate effect acquired an average error rate of 7.823% whereas Oliver-Pharr method acquired an average error rate of 6.355%, both with comparison against NanoTest<sup>TM</sup> machine. The experiments have been performed using same materials; i.e. brass, aluminium, copper and mild steel.

*Index Terms*— Nanoindentation, Oliver-Pharr, Joslin-Oliver, Load-depth curve.

## I. INTRODUCTION

Hardness indentation test is an essential need towards industry and also to identify the characteristic of the material. Hardness is a material property that defines the resistance between plastic penetration, indentation and deformation. A problem indentation process usually faced is its size scale limitation whereby the size of the indenter exceeds the test material size. Hence, indentation test is designed in different scales from micro to nano with the application of different diamond tip geometry such as Brinell, Vickers and Berkovich. This reduces the risk of material spalling whereby the indentation is performed at small depth

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and small load. The project is designed with a Nano Indenter II head assembly which includes a displacement sensor, voice coil actuator and diamond Berkovich indenter tip. The designed software implements the preferred Oliver-Pharr method while assuming the material surface is smooth together with Joslin-Oliver method which is able to eliminate substrate effect during indentation test.

Oliver-Pharr method identifies the hardness properties based on area and depth impression acquired during indentation process. This is a popular method as smart vision is not required during the acquisition of the area of indent. However the estimation of contact area based on loaddisplacement curve can be inaccurate and subsequently affect the calculation of hardness and modulus. Hence, according to Seung et al. [1], indenting the tip within a depth of  $\sim 10\%$  of the thin film can resolve this issue which leads to another problem that is the  $\sim 10\%$  depth may not be as accurate caused by the specimen surface roughness. The solution to this problem is then referred to Han-Saha hardness which is based on the reduced modulus model by Saha and Nix,  $E_r(Saha)$  [2], and used by Han et al. [3] with both based on the assumption of King's flat punch model that is applied on the indenter tip that allows penetration through the film. Joslin-Oliver method in another case, applies to elastically mismatched film whereby it identifies how contact stiffness or modulus is being affected by the increased of indentation depth. The model is as followed:

$$H_{JO} = \frac{P}{A_c} = \beta_i^2 \frac{4}{\pi} \frac{P}{S^2} E_r^2$$
(1)

Where P is load, Ac represents contact area,  $\beta_i$  carries the Berkovich constant value of 1.034, S is stiffness and Er is the reduced modulus.

#### II. METHODOLOGY

#### A. Oliver-Pharr Method

The designed nanoindentation system is required to measure the hardness and elastic modulus without the need of image capturing, hence the analytical method introduced by Oliver and Pharr is being reviewed due to its adoptability in characterizing small-scale mechanical behaviour [4]. The mechanical properties can be identified based on the loaddisplacement data acquired at single loading and unloading cycle. However, usual elasticity-based analytic measurement

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occurred at the unloading curve which represents the material elastic recovery. The relationship between the depth, h and load, P at the unloading curve is represented as shown [5]:

$$P = \alpha \left( h - h_f \right)^m \tag{2}$$

Where  $\propto$  represents the geometric constants and m represents the theoretical constant value of contact geometry [5]. Parameter maximum displacement, h and final displacement, h<sub>f</sub> can be identified through the P-h curve. Elastic deformation between the indenter and test piece is evaluated as the reduced modulus,

$$E_r = \frac{\sqrt{\pi}}{2\beta} \frac{S}{\sqrt{A}} \tag{3}$$

S is obtainable through the unloading of P-h curve whereas contact area, A is calculated through the area function which depends on the indenter scale and geometry [6]. In order to fulfil contact area calculation, the actual indented depth is identified through,

$$h_c = h_t - h_s \tag{4}$$

where  $h_s$  is the elastic displacement and can be determined by using Eq (5):

$$h_{S} = \frac{\varepsilon}{S} P_{\max}$$
(5)

 $h_s$  can be defined as sink-in of material surface at contact perimeter [5]. The behaviour of the contact depth during unloading and its parameters characterization of contact geometry is as illustrated in Fig. **1**. Furthermore,  $\varepsilon$  is given as a theoretical value of 0.75 for Berkovich indenter that corresponds to m that varies according to the geometry tip. Lastly, the hardness of the material can be acquired using the equation given by:





Fig. 1. Unloading parameters of contact geometry.

## B. Oliver-Pharr Algorithm Program Flow

During indentation process, it is required to identify the specimen hardness and reduced modulus without the need of visual instruments such as scanning electron microscope (SEM) and atomic force microscopy (AFM), the task needs to be done purely based on the displacement data collected during indentation process. Hence the software of the system is designed to identify each variable required to perform the calculation for hardness by using Eqn (6) and reduced modulus by using Eqn (3).

The program is able to extract specific values from the load-depth curve to identify the indent area with the application of the following formula:

$$A = f(h_c) = 24.56h_c^2 + 0.562h_c + 0.003216$$
(7)

In this case, C is assumed to be zero, hence acquiring  $h_c$  contact depth needs to be estimated based on:

$$h_c = h_t - \frac{\varepsilon}{S} P_{\max}$$
(8)

with  $\varepsilon = 0.75$  for a paraboloid tip due to the tip selection is diamond Berkovich. Though the contact depth is needed for the area measurement, stiffness, S is also required to fulfil the calculation as well. Therefore the formula as shown below is prioritized before proceeding to hardness and reduced modulus measurement.

$$S = \frac{dP}{dh} = \frac{P_{\text{max}}}{h_t - h_f} \tag{9}$$

The following stiffness calculation is based on the upper 30% - 40% of unloading curve which stated by Qianhua Kan et al that this allows a more effective reduced modulus calculation as compared to 25% - 50% suggested by Hay and Pharr [4, 7].

## C. DAQ Hardware and Software

National Instruments myRIO-1900 is chosen as the data acquisition hardware and controlling system for the project due to its suitable output voltage required for supplying the actuator and controlling the load. In addition, it also consists of analogue input pin that allows displacement signal acquisition. Moreover, the device is small in dimension which meets the portability requirement and it is programmed by using NI LabVIEW DAQ software [8].

National Instruments LabVIEW is an analytical design and measurement software which consist of controlling capabilities and also runs in graphical programming interface that allows ease of signal monitoring and processing. Furthermore, the software platform is designed to control as a stand-alone instrument which is capable at running in automated real-time system [9].

## D. Voice Coil Actuator

The Voice coil actuator is designed with a permanent magnet and coil winding as a conductor which allows limited motion according to the current applied to the conductive coil which is proportional to the output force produced. Voice coil actuator can be designed in linear motion and the force can be varied according to input current as well. The selected voice coil actuator is integrated in the head assembly together with a build in capacitive displacement sensor.

Voice coil actuator holds a simple construction with no gears included as compared to motor actuator. The force produced by the actuator varies directly from the current supplied due to the constant electrical resistance of the coil. However, during indentation process, higher force is required to act against the specimen hence the actuator itself draws larger current to oppose the indentation force [10].

## III. EXPERIMENT AND RESULTS

The nanoindentation system designed in this project consists of a main controller, indenter head assembly, aluminium frame and stainless steel adjustable stage. Fig. 2 illustrates the hardware of the nanoindentation system.



The main controller of the system used is National Instruments myRIO-1900 data acquisition hardware which allows communication between LabVIEW that performs the I/O data acquisition from the indenter head assembly. On the other hand, the head assembly contains a voice coiled actuator which produces current in the coil assembly when load coaxial port is connected and controlled within +5 to -5 V. This causes the coil to be either pull towards the internal permanent magnet or push away, depending on the load voltage.

Furthermore, the head assembly also consists of a displacement sensor which reads the movement of the actuator tip by a 3 plate capacitive gauge. The measurement of the sensor requires a positive and negative voltage of 15 V which is connected to the female D-sub connector. Another

important feature of the system is the indentation stage where its height is adjustable according to the size of the specimen.

## A. Brinell Hardness Test

Average diameter (mm)

nell Hardness (BHN)

20000

The material properties recorded from the following Brinell experiment has been divided into two different loading for comparison purpose as shown in TABLE 1 and TABLE 2.

 TABLE 1

 BRINELL HARDNESS MEASUREMENT WITH 30kN LOAD

 rre (N)
 Test Piece
 Copper
 Aluminum
 Brass
 Mild Steel

Eorco (NI)	restricte		oppe						01000		initia oteet		
Force (N)	Trials	1	2	3	1	2	3	1	2	3	1	2	3
	Diameter of indentation (mm)	6.5	6.7	6.4	6.4	6.3	6.7	6.3	6.3	6.0	5.5	5.6	5.6
30000	Average diameter (mm)		6.533		6.467		6.200		5.567				
	Brinell Hardness (BHN)	80.167		7	82.094		90.413		115.056		6		

		T.	ABI	JE 2	2								
1	BRINELL HARDNESS N	ME/	٩SU	RE	ME	NT	WI	TH 2	20k1	N LO	DAI	)	
Form (NI)	Test Piece	0	oppe	r	A	umini	um		Brass		M	ild Ste	el
Force (IN)	Trials	1	2	3	1	2	3	1	2	3	1	2	3
	Diameter of indentation (mm)	5.2	5.4	5.2	5.6	5.6	5.7	4.9	5.0	5.0	4.0	4.1	3.8

4.967

98.314

3.967

158.260

5.633

74,715

5.267

86 596

30 kN loading force shows a promising result whereby the BHN acquired are within the range of typical soft brass and mild steel material hardness. Mild steel holds the highest BHN while copper holds the least, due to brass is a composite comprised with copper and zinc which makes it harder compared to copper. As for the 20 kN loading experiment, the results are slightly off and this may be caused by several predicted factors:

- 1. Surface roughness affects the Brinell hardness test.
- 2. Indentation taken too close to the edge causes side bulging that affects the results.
- 3. The indentation does not meet the required depth of impression.

From the 3 factors listed above, it is more likely that the inaccurate results is due to the depth of impression during indentation. Hence, the indentation depth needed is estimated by using the following formula:

$$t = \frac{1}{2} \left[ D - \sqrt{D^2 - d^2} \right]$$
(10)

$$t = \frac{P}{BHN \times \pi \times D} \tag{11}$$

Equation (10) is used to calculate the depth by using existing diameter of indentation whereas (11) is used to calculate based on the 30kN hardness results under 20kN calculation to predict the required indentation depth to achieve approximate similar hardness. The error rate between the achieved depth of impression and estimated depth of impression under 20kN has been estimated as shown in TABLE 3

TABLE 3 DEPTH OF IMPRESSION MEASUREMENT

Force (N)	Test Piece	Copper	Aluminum	Brass	Mild Steel				
30000	Depth of impression (mm)	1.215	1.186	1.077	0.846				
20000	Depth of impression (mm)	0.750	0.869	0.660	0.410				
20000	Expected depth from 20kN load	0.810	0.791	0.718	0.564				
	Error rate (%)	7.425	9,876	8.036	27,299				

illustrated in Fig. 4.

ht

indenter

The results acquired in Table show that mild steel carries the highest error rate of 27.299%. Two hypotheses have been made from this occurrence: 1. The high elasticity of mild steel causing the dwell time of 15 seconds is too short to build the expected impression depth/plastic depth (h<sub>f</sub>), causing a high contact depth (h<sub>c</sub>) to occur during the unloading of the tip. 2. The supplied load of 20 kN is not enough to penetrate the material to its desired impression depth. In order to acquire more accurate results when using 30 kN load, the dwell time needs to be taken into consideration where the depth of impression varies whenever the dwell time extends.

This experiment is able to conclude that loading timing profile holds an important role during a hardness indentation process and this may affect the accuracy of the final hardness results. Therefore, by designing a user friendly loading timing profile, it increases the flexibility to vary the impression depth during indentation process.

### B. Simulation – Oliver-Pharr

The simulation conducted utilizes data extracted from NanoTest<sup>TM</sup> results acquired from similar materials. During NanoTest<sup>TM</sup> nanoindentation test, a load-displacement signal is acquired and the values obtained for the simulation are as shown in Fig. 3. The circled points in the graph representing the maximum load, maximum depth and plastic depth.



Fig. 3. Load-displacement curve - data extraction



The Oliver-Pharr hardness derivation is performed as shown:

loaded Fig. 4. Highlighted variable needed for hardness calculation

# C. Simulation – Joslin-Oliver

 $h_c$ 

The Joslin-Oliver method holds an advantage to nanoindentation where it eliminate substrate effect while assuming the indentation depth does not affecting the reduced modulus, Er, whereby it is at fix state. The hardness can be calculated based on the formula as shown:

Where Hop represents Oliver-Pharr hardness, h<sub>f</sub> is the plastic

depth, ht is the maximum depth and Pmax is the maximum

load. During Oliver-Pharr measurement, the material surface

is assumed to be smooth and does not pile-up or sink-in during

indentation. The variables for hardness calculation are as

h

Pmax

initial surface

unloaded

$$H_{JO} = \frac{4\beta^2}{\pi} \times \frac{P_{\text{max}}}{S^2} \times E_r^2$$
(13)

Where S, stiffness can be acquired based on Eqn (9), reduced modulus,  $E_r$  from Eqn (3) and  $\beta = 1.034$  constant value for Berkovich indenter tip.

TABLE 4 shows the recorded hardness results for the four materials where copper with 1.508 GPa, aluminium with the lowest GPa of 1.323, brass with the second highest GPa of 2.879 and lastly, mild steel with the highest hardness GPa average value of 3.105. In summary, the hardness sequence for the material in this experiment is proven correct, but slightly higher as compared to Oliver-Pharr method due to the assumption of reduced modulus being constant.

TABLE 4									
JOSLIN-OL	IVER METHOD	4 MATE	RIALS S	IMULAT	ION RES	SULTS			
	Reduced Modulus	9.250	8.892	9.115	9.533	9.344			
	Hardness (GPa)	1.595	1.409	1.466	1.564	1.505			
Copper	Hardness (HV)	162.620	143.636	149.464	159.496	153.436			
	Average Hardness (GPa)	1.508							
	Average Hardness (HV)	153.730							
	Reduced Modulus	4.811	4.687	4.763	4.878	4.791			
	Hardness (GPa)	1.358	1.294	1.267	1.307	1.390			
Aluminum	Hardness (HV)	138.465	131.928	129.210	133.232	141.775			
	Average Hardness (GPa)	1.323							
	Average Hardness (HV)		134.922						
	Reduced Modulus	6.962	6.503	6.779	6.779				
	Hardness (GPa)	2.721	2.776	2.785	3.234				
Brass	Hardness (HV)	277.449	283.042	283.992	329.787	-			
	Average Hardness (GPa)								
	Average Hardness (HV)		293	293.567					
	Reduced Modulus	14.293	12.633	15.240	5.632	14.718			
	Hardness (GPa)	3.133	3.435	3.107	2.952	2.900			
Mild steel	Hardness (HV)	319.515	350.243	316.865	300.990	295.675			
	Average Hardness (GPa)			3.105					
	Average Hardness (HV)			316.658					

TADLE 4

## D. Simulation – LabVIEW Interface

Each experiment is performed using LabVIEW software where the program is able to control the pins of myRIO-1900 DAQ directly while acquiring data for material hardness measurement. Fig. **5** shows the first page of the program where it allows user to control the load timing with the objective to increase depth while reducing its load. It also simultaneously acquire the displacement signals as well.



Fig. 5. LabVIEW - Load timing profile settings and displacement reading

Fig. 6 provides another option to the user by inserting total depth, plastic depth and maximum load parameters for hardness calculation through Oliver-Pharr and Joslin-Oliver method. This feature allows flexibility towards comparison between different algorithms throughout the experiment.



Fig. 6. LabVIEW - Oliver-Pharr & Joslin-Oliver hardness acquisition

### IV. DISCUSSIONS AND COMPARISONS

The aim of the comparison is to acquire minimal error rate nanoindentation technique to be implemented in the nanoindentation system.



As shown in Fig. 7, the first comparison includes Brinell test, NanoTest<sup>TM</sup>, simulation test based on Oliver-Pharr method and Joslin-Oliver method. The results is tabulated in TABLE 5. Due to Brinell is not under nanoindentation category, its hardness values are converted to Mohs for comparison purpose. Therefore, copper held a Mohs hardness value at the range of 2 - 2.5 whereas Oliver-Pharr hardness is at the range of 2.7 - 2.8. Joslin-Oliver consists of a larger range compared to Brinell hardness which may be caused by Brinell test inconsistency of reduced modulus value. Furthermore, mild steel for Brinell test holds a Mohs value range of 2.7 - 2.8, which may cause reading error against copper readings from Oliver-Pharr method. Hence, Mohs hardness conversion value is not suitable to perform between indentation comparison and nanoindentation hardness.

 TABLE 5

 COMPARISON BETWEEN BRINELL, NanoTest™ AND LabVIEW

 SOFTWARE SIMULATION HARDNESS RESULTS

		Tardicis,							
	Comparison	Indentation		Nanoindentation					
		Brinell Test	NanoTest TM	Simulation (Oliver-Pharr)	Simulation (Joslin-Oliver)				
	Copper	80.167	140.249	134.486	153.730				
Matoriala	Aluminum	82.094	126.567	118.032	134.922				
Waterials	Brass	90.413	267.685	245.452	293.567				
	Ball d Cho al	115.055	202.004	077.010	216 652				

TABLE 6 compared between NanoTest<sup>TM</sup> machine hardness estimation and the simulation software based Oliver-Pharr method, with an average error rate of 6.355% however the hardness sequence of the material is still correct.

 TABLE 6

 NanoTest<sup>TM</sup> VS SIMULATION (OLIVER-PHARR) HARDNESS

		Comparison	Nan	Error Data (%)		
		Comparison	NanoTest TM	Simulation (Oliver-Pharr)	EITOI Nate (%)	
		Copper	140.249	134.222	4.297	
	Matorials	Aluminum	126.567	117.645	7.049	
	Waterials	Brass	267.685	244.494	8.664	
		Mild Steel	292.864	277.018	5.411	
				Average Error Rate (%)	6.355	

As for the comparison between NanoTest<sup>TM</sup> and Joslin-Oliver method, the machine identifies the hardness by accepting every factor that may affect the results including the substrate effect, while Joslin-Oliver eliminates the substrate effect by assuming the reduced modulus is constant. Therefore the final calculation results in an error rate of 7.823%, which is higher than Oliver-Pharr method as shown in TABLE 7.

TABLE 7

NanoTest<sup>TM</sup> VS SIMULATION (JOSLIN-OLIVER) HARDNESS

	Comparison	Nan	Error Data (%)	
	Comparison	NanoTest TM	Simulation (Joslin-Oliver)	EITOI Nate (70)
	Copper	140.249	153.730	8.769
Matorials	Aluminum	126.567	134.922	6.193
Waterials	Brass	267.685	293.567	8.816
	Mild Steel	292.864	316.658	7.514
			Average Error Rate (%)	7.823

From the Brinell test, it is observable that the time of indentation defines the depth and area of the indent impression which may affect the hardness results. On the other hand, nanoindentation aims to reduce material used and avoid spalling of material. In this case, Brinell test is able to reduce its indentation depth by reducing the load while increasing the dwell time. This is to form a better impression depth in order to achieve a desired area to measure the hardness of the material. This not only reduces the area of indent but also allows Brinell test to be performed on thinner materials. Furthermore, the nanoindentation tests and its comparison are able to conclude that there are still many parameters needs to be considered while measuring the hardness at nano scale. These factors include gravity, material's surface roughness and environment sound frequency which may cause vibrations.

### V. CONCLUSION

The project concludes the essentiality of acquiring a smart vision scope which allows the user to identify substrate error or indentation error which might affect the results. It is advisable to construct a nanoindentation system using separated components due to the sensitivity of actuator and displacement sensor. Avoid using glass made capacitive displacement sensor that can be broken easily. Instead, opt for laser sensor which is able to produce the desired output. While constructing the nanoindentation system, precision X-Y axis motorized stage is able to allow the indentation to take place with small range in between that saves space and material which meets the main intension of a nanoindentation system. As to avoid vibration emitted from the ground, magnetic floating table can be implemented to reduce the error rate during nanoindentation process. Last but not least, the project experiment may also concludes that the criticalness of dwell time during indentation will affect the depth of impression and subsequently resulting in false measurement of the hardness of the materials.

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