Enhanced Sensitivity of Humidity Sensor Prepared using Vertically Aligned V-doped TiO₂ Nanorods Array

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Abstract—This article reports on the performance of vanadium-doped TiO₂ nanorod arrays (TNA)-based humidity sensor. The sensing material was prepared using a facile solution immersion method. Humidity sensor was fabricated using the prepared thin films and was tested using humidity chamber and a current measuring circuit, built with microcontroller. Improvement to the sensitivity was observed for the V-doped TNA sensor with sensitivity value of 15.0 compared to 9.9 for the undoped TNA sensor. The increased performance is attributed to the enlargement of nanorods diameter, improvement of crystallinity and the increase in oxygen vacancy defect sites of the V-doped sample.

Index Terms—humidity sensor, nanorods, titanium dioxide, vanadium

I. INTRODUCTION

The amount of water molecules in the air is often referred to as the humidity. It is an important physical parameter as it can give significant effect to the human's everyday life. Accurate measurement of this quantity is needed in agricultural [1], food manufacturing [2], health monitoring [3, 4] and weather forecasting. To achieved this, various kinds of humidity sensors have been produced using materials such as polymers [4] and metal oxides.

Metal oxides have the advantage over polymers due to the robust nature of the material, enabling it to be employed in harsh condition such as in high temperature and chemically challenging environment. One of the proven effective metal oxides material is titanium dioxide (TiO₂). Armed with its high hydrophilicity and highly tunable nanostructure, TiO_2 has become one of the preferred metal oxides for humidity sensor application. While TiO_2 could exist in three crystalline phases of rutile (tetragonal), anatase (tetragonal) and brookite (orthorhombic) [5], most of the reported humidity sensor utilizes rutile or anatase phase TiO_2 .

A good humidity sensor should have high sensitivity, fast response/recovery time, and high stability [6]. Various efforts have been carried out by researchers to further improve the performance of TiO_2 -based humidity sensor. Fabrication of unique nanostructures such as nanosheets [7], nanoflowers [8], nanotubes [9], nanofibers [10], nanowires [11], and nanorods [12] have proven beneficial in enhancing the humidity sensor performance. Since the detection of humidity water molecules correlates with the adsorption of water molecules on the sensor surface [13], increment in the surface area are often cited to be the reason for the improvement.

In the case of one-dimensional (1D) nanostructures such as nanorods and nanowires, the direct pathway for charge carrier transport is said to be responsible for the enhancement in sensor performance. Having oriented structures ease the transport of electron if compared to other non-oriented nanostructures [5]. Meanwhile, there are also attempts by other groups to dope TiO₂ with other elements such as cobalt [14], copper [15], and tin [16] to increase the sensitivity of humidity sensor. The purpose was to increase the carrier concentration by donation of extra charge carrier through substitutional doping [17]. Besides that, doping could also increase the oxygen vacancies sites. This type of defect helps the humidity detection by increasing water molecules adsorption sites [18]. A density-functional theory (DFT) calculation by Farzaneh et al. [10] showed that adsorption energy of water molecules to rutile TiO₂ increased to 87% with the addition of ruthenium dopant.

In this work, we aim to combine these two approaches by synthesizing V-doped TiO_2 nanorod arrays (TNA) using a facile solution immersion method and to fabricate humidity sensor using the prepared thin films. V was chosen as the dopant as it has similar ionic radius as Ti ion [19] which would allow for smooth incorporation of the dopant. It has also been proven to improve the photocatalytic performance of TiO₂. However, to the best of our knowledge, there have

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not been any report yet on the fabrication of V-doped TNAbased humidity sensor.

II. METHODOLOGY

A. TNA Preparation

The TNA was prepared using method developed by Yusoff et al. [20]. Briefly, a solution was prepared by mixing deionized (DI) water with hydrochloric acid (37%, Merck) at 1:1 ratio in Schott bottles. 0.07M of tetrabutyl titanate (97%, Sigma-Aldrich) was added to the solution as the precursor. For V-doped sample, vanadium chloride was further added to the solution as the dopant source. The solution was stirred rigorously for 50 minutes before flourine-doped tin oxide (FTO)-coated glass substrate was positioned inside the bottle with the FTO layer facing upward. To allow for the growth of the nanorods, the bottle was placed inside an oven set at 150°C. After 3 hours, the substrate was carefully taken out and rinsed with DI water before being blow-dried with nitrogen gas.

B. Characterization

The thin films' morphology was observed using fieldemission scanning electron microscopy (FESEM, JEOL JSM-7600F) and its structural properties was characterized using X-ray diffractometer (XRD, PANalytical X'pert PRO). For humidity sensor measurement, silver (Ag) metal contact was deposited on top of the thin films using thermal evaporator (ULVAC). The humidity sensor measurement was done inside a humidity chamber (ESPEC-SH261). Measurement system was built using Arduino ATmega2560 microcontroller, 100Ω resistor and a homemade acrylic stand as shown in Fig 1.



Fig. 1. Circuit connection of humidity measurement system.

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III. RESULT AND DISCUSSION

A. Structural Properties

Fig. 2 shows the FESEM top view images of the prepared thin films. Both samples showed uniformed and vertically aligned nanorod arrays. The average nanorod diameter was measured at around 100 nm and 200 nm for the undoped and V-doped TiO₂ respectively, suggesting that the introduction of V increases the grain size of TNA. This result is in agreement with the report by Faia et al. [20]. It appears that the addition of V promotes the crystal growth of TiO₂. As a result, the nanorods in the V-doped TNA sample are more closely packed compared to the undoped TNA.



Fig. 2. FESEM top-view images of: (a) undoped TNA, (b) V-doped TNA at 30k times magnification.

The FESEM cross sectional view of the samples is shown in Fig. 3. From the figure, the average TNA thickness is estimated at 3.05 μ m and 2.17 μ m for the undoped and V-doped sample respectively. From this result, it can be concluded that the addition of V appears to increase the growth of the nanorod structural laterally while inhibiting its vertical growth. The FTO layer also clearly visible at the bottom of TNA. This FTO layer is essential as the seed layer for the vertical growth of TNA. Since it shares the same tetragonal structures as TiO_2 , it could provide a suitable nucleation sites for the growth of TiO_2 crystals.



Fig. 3. FESEM cross-sectional view images of: (a) undoped TNA, and (b) V-doped TNA at 15k times magnification.

XRD spectra of the samples is shown in Fig. 4. Peaks which correspond to (110), (101), and (002) planes were observed for both of the samples indicating the presence of rutile phase TiO₂. The absence of vanadium oxide's characteristic peaks suggests that either the V ions are incorporated in TiO₂ lattice or vanadium oxide is too small to be detected by the XRD [22]. The V-doped TNA showed higher peak intensity for all TiO₂ related peaks which suggests better crystallinity than the undoped TNA. This finding is supported by Liu et al. [22] which suggested that the doping of V in TiO₂ lattice improved the growth of TiO₂ nanoparticles and is in line with the previous FESEM result. The crystallite size of the samples was calculated using Scherrer's equation as follows:

$$D = \frac{K\lambda}{\beta \cos\theta}$$
(2)

where **K** represents a constant, given as 0.9, λ is the X-ray's wavelength (1.5418Å), **\theta** denotes the diffraction angle while **\beta** refers to full-width at half maximum (FWHM) of the peak. From the calculation of the peak which corresponds to (110) plane, the undoped TNA recorded crystallite size of 40.4 nm while the V-doped TNA crystallite size is 196.4 nm. This significant enlargement of the TiO₂ crystallite size with the addition of V, further confirms the crystal growth promotion caused by V doping.



Fig. 4. XRD spectra of the undoped and V-doped TNA.

B. Humidity Sensor

In order to gauge the performance of the humidity sensor, each device was placed in a humidity chamber and the humidity level was elevated from 40% to 90% Relative Humidity (RH), before going back to 40% RH. The output signal value from the microcontroller was recorded at every 10% RH interval. Fig. 5 shows the humidity sensing response of the undoped and V-doped humidity sensor. Both samples depict sharp increase in signal intensity when the humidity was increased from 40% to 90% RH, before going back down when the humidity was reduced to 40% RH, suggesting that both type of sensor are able to sense the changes in humidity. The sensitivity value, **S** was calculated using the following equation

$$S = \frac{R_a}{R_{rh}} = \frac{Output \ signal \ at \ 90\% \ RH}{Output \ signal \ at \ 40\% \ RH}$$
(1)

From the calculation, the undoped TNA sample recorded *S* value of 9.9 while the V-doped TNA at 15.0.

The improvement to the S value for the V-doped TNA can be attributed to several factors. It is postulated that the

main contributing factor is the increased size of the Vdoped TNA. The mechanism of humidity detection for a resistive type humidity sensor is the modulation of the film's resistance with the changes in relative humidity. When water molecules are absorbed on the thin film's surface, more charge carrier in the form of electron or ions are available in the thin films thus increasing the thin film's conductivity [23]. As can be seen from the FESEM images, the V-doped TNA are having larger nanorods diameter resulting in a more condense packed nanorods array. It is believed that the closely connected nanorods array will facilitate the movement of the charge carrier, resulting in higher sensitivity. Even though some other researchers have argued that smaller diameter nanorods have the advantage of higher surface area compared to the larger ones, it is believe that in our case, the effect of better electron transportation surpasses this effect.

Furthermore, as mentioned earlier in the XRD result section, the crystallinity of the V-doped TNA is better than the undoped TNA, which is evident by its higher peak intensity. Since better crystallinity is often associated with increased conductivity [24], it is conceivable that this improvement also plays a role in enhancing the sensitivity of the V-doped TNA sensor.

Another possible reason for the improvement seen in Vdoped TNA sensor sensitivity value is through the increase in defect sites particularly oxygen vacancies, usually associated with metal doping of TiO₂. This defect sites would increase the amount of trapped electron which will then be liberated by the adsorption of water molecules [25].



Fig. 5. Humidity sensing response of the undoped and V-doped humidity sensor.

IV. CONCLUSION

V-doping of TNA was achieved through a facile solution immersion method and its structural properties were investigated before being tested for humidity sensor application. Compared to the undoped TNA, the V-doped device showed improvement in term of sensitivity. This improvement is attributed to the enlargement of nanorods diameter, enhancement in the crystallinity and the increase in oxygen vacancy defect sites of the V-doped sample. These findings could help improve the performance of TiO₂-based resistive type humidity sensor.

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